

**Zeitschrift:** Schweizerische mineralogische und petrographische Mitteilungen = Bulletin suisse de minéralogie et pétrographie  
**Band:** 44 (1964)  
**Heft:** 1  
  
**Artikel:** Detailed crystal structure of rhombohedral MoS<sub>2</sub> and systematic deduction of possible polytypes of molybdenite  
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**DOI:** <https://doi.org/10.5169/seals-34328>

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# Detailed Crystal Structure of Rhombohedral MoS<sub>2</sub> and Systematic Deduction of Possible Polytypes of Molybdenite<sup>1)</sup>

By *Y. Takéuchi* and *W. Nowacki* (Bern)<sup>2)</sup>

With 9 figures in the text and 3 tables

## Abstract

The rhombohedral modification of molybdenite which was recently found in Binnatal (Switzerland) has the lattice constants:  $a_0 = 3,166 \pm 0,005$  Å and  $c_0 = 18,41 \pm 0,02$  Å. The space group is  $C_{3v}^5$ — $R3m$ . There are three formula units in the cell. The crystal structure was investigated by Fourier methods and a trigonal prismatic coordination of Mo atoms was established. The Mo-S distances are  $2,41 \pm 0,02$  Å.

Possible polytypes with simple structures were theoretically derived. The results give one rhombohedral, two hexagonal and one trigonal structures. They are respectively designated by  $3R$ ,  $2H_1$ ,  $2H_2$  and  $2T$ . A comment on the stacking operations of the  $2T$  structure is given.

## 1. Introduction

The crystal structure of molybdenite, MoS<sub>2</sub>, was first investigated by DICKINSON and PAULING (1923). It is hexagonal and the structure is characterized by MoS<sub>2</sub> layers in which the Mo atoms have trigonal prismatic coordination of six sulfur atoms. In 1957, BELL and HERFERT artificially obtained a rhombohedral modification of MoS<sub>2</sub> and they proposed, by qualitative study of (00 $l$ ) reflexions, a similar layer structure but the Mo atoms having octahedral coordination. JELLINEK et al. (1960) studied synthetic rhombohedral modifications of MoS<sub>2</sub> and pointed out that the coordination polyhedra of the Mo atoms are trigonal

<sup>1)</sup> Contr. no. 142; paper no. 9 on sulfosalts and sulfides from Binnatal (Switzerland).

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prisms. It has been reported that another possible new modification of  $\text{MoS}_2$  was obtained by ZELIKMAN et al. (1961), though the structure of the modification has not been established.

In Nature, since the work by DICKINSON and PAULING, molybdenite seemed to exist only in the hexagonal form until quite recently GRAESER (1963) found by X-rays a non-hexagonal form of natural molybdenite in Binnatal and TRAILL (1963) identified the same modification, found in Con mine, Yellowknife, District of Mackenzie (Canada), as rhombohedral.<sup>3)</sup> GRAESER kindly gave us his material for further investigation. In view of these circumstances, it seemed worthwhile to make a theoretical derivation of possible polytypes of this particular layer structure and, on the other hand, to establish the detailed configuration of the  $\text{MoS}_2$ -layers. This investigation was undertaken as part 9 of researches on sulfides and sulfosalts from Binnatal (Part 8, see NOWACKI et al., 1963, 1964).

## 2. Experimental

The crystals of rhombohedral molybdenite are well developed and a suitable specimen for X-ray work was successfully prepared. It is practically free from distortion and has a cross section  $0,15 \times 0,02$  mm. To minimize the effect of absorption,  $\text{MoK}\alpha$  radiation ( $\lambda = 0,7107 \text{ \AA}$ ) was used with Zr metal filter throughout this work. For  $\text{MoK}\alpha$ , the linear absorption coefficient of  $\text{MoS}_2$  was found to be as low as  $73,5 \text{ cm}^{-1}$  (calculated from  $\mu_a$ ). Lattice constants found in precession photographs are:

$$a_0 = 3,166 \pm 0,005 \text{ \AA}, \quad c_0 = 18,41 \pm 0,02 \text{ \AA}$$

The possible space group  $C_{3v}^5 - R3m$  was confirmed by the later stage of structure analysis. There are three formula units in the hexagonal cell. These lattice constants give the calculated density  $4,99 \pm 0,01 \text{ g/cm}^{-3}$  which should be compared with the density  $5,022 \pm 0,002 \text{ g/cm}^{-3}$  for the synthetic rhombohedral  $\text{MoS}_2$  (BELL and HERFERT, 1957). By a chemical treatment, it has been proved that the molybdenite is practically pure  $\text{MoS}_2$  (GRAESER, 1963).

The intensities of 104 ( $h0l$ ) reflections were measured from the integrated Weissenberg diagrams with the help of a Joyce-Loebl-densitometer. Transmission factors for each reflection were graphically estimated (TAKÉUCHI, 1958), by assuming a rectangular cross section.

<sup>3)</sup> Recently A. H. CLARK [Miner. Soc., Notice No. 126, Meeting on 30 January 1964 in London] gave a paper on "Rhombohedral molybdenite from Minas da Panasqueira, Beira Baixa, Portugal" (Occurrence in greisenized granite).

### 3. Analysis

There are three Mo and six S atoms in the hexagonal cell. It is, therefore, expected that they are distributed over the  $3(a)$  positions of the space group. The problem is then to determine three  $z$ -parameters of the atoms. The zero-layer reflections about one of the hexagonal  $a$ -axes were indexed based upon an oblique cell as shown in Fig. 1. The cell has the plane symmetry  $p_1$ . Calculations of structure factors and Fourier syntheses were performed on the basis of this oblique cell. The number of parameters were, then, increased to six because of lower symmetry of the cell. Indices of reflections were finally transformed to those of hexagonal axes.

A Patterson projection along one of the  $a$ -axes was first prepared by the aid of the von Eller machine. This is shown in Fig. 2. From this vector-map it is immediately seen that the Mo-atoms should have a trigonal prismatic coordination of sulfur atoms. The pairs of peaks

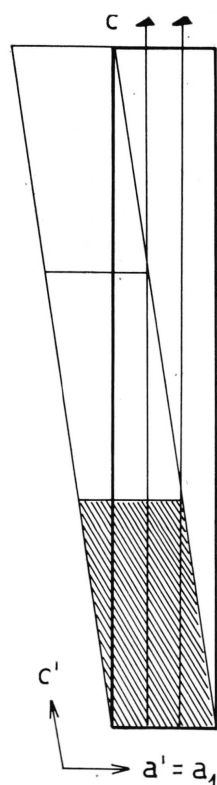


Fig. 1. The relation of the oblique cell (shaded) and rhombohedral cell as projected along the  $a_2$ -axis. The rhombohedral cell is outlined by heavy lines. Three fold axes are indicated.

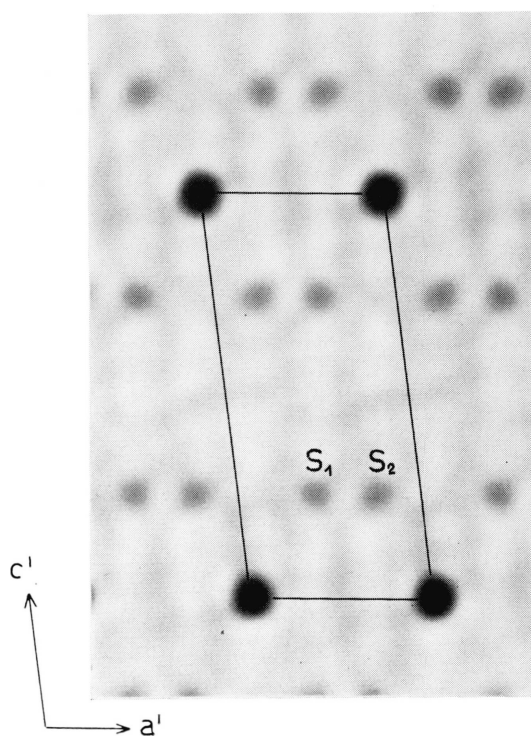


Fig. 2. Patterson projection of rhombohedral MoS<sub>2</sub> along  $a_2$ -axis.

Table 1. Comparison of  $F_0$  and  $F_c$  of rhombohedral molybdenite

$h'$	$l'$	$h$	$l$	$F_0$	$F_c$	$A_c$	$B_c$	$h'$	$l'$	$h$	$l$	$F_0$	$F_c$	$A_c$	$B_c$
0	1	0	3	33,7	38,2	38,2	0	$\bar{1}$	1	$\bar{1}$	2	28,2	29,1	27,9	8,5
	2		6	8,4	12,1	12,1	0		2		5	39,2	41,6	38,8	-14,8
	3		9	30,8	31,5	31,5	0		3		8	31,1	30,8	30,2	-5,7
	4		12	39,9	40,4	40,4	0		4		11	20,4	20,7	16,5	12,5
	5		15	17,1	18,4	18,4	0		5		14	19,4	19,2	18,8	3,7
	6		18	5,0	6,7	6,7	0		6		17	24,0	26,6	24,5	-10,3
	7		21	11,4	19,7	19,7	0		7		20	17,8	17,7	17,5	-2,1
	8		24	20,2	22,8	22,8	0		8		23	12,9	12,5	9,7	7,8
	9		27	12,0	9,2	9,2	0		9		26	12,4	11,7	11,7	0,9
1	0	1	1	27,1	28,8	24,1	-15,9		10		29	14,8	14,8	13,7	5,6
	1		4	36,6	37,9	36,7	9,3		11		32	7,8	9,1	9,1	0,2
	2		7	35,9	37,4	35,2	12,4	$\bar{2}$	1	$\bar{2}$	1	19,7	20,1	16,7	-11,2
	3		10	18,1	21,6	19,9	-8,5		2		4	25,6	27,7	26,8	7,1
	4		13	16,8	18,9	16,4	-9,4		3		7	26,9	28,9	27,2	9,9
	5		16	22,8	24,9	23,7	7,7		4		10	19,5	17,3	15,8	-7,2
	6		19	21,0	22,1	21,0	6,8		5		13	15,8	15,8	13,7	-7,9
	7		22	12,7	12,9	11,2	-6,3		6		16	21,2	21,4	20,3	6,6
	8		25	10,0	11,2	10,2	-4,5		7		19	19,9	19,0	18,1	5,7
	9		28	12,2	14,6	13,8	4,8		8		22	12,4	11,2	9,8	-5,4
10			31	12,5	11,4	11,1	2,8		9		25	12,4	9,9	9,1	-3,8
									10		28	15,3	13,0	12,3	4,2
									11		31	11,3	10,2	9,9	2,4
2	0	2	2	20,1	20,4	19,4	6,2	$\bar{3}$	1	$\bar{3}$	0	27,2	28,7	28,7	-0,0
	1		5	29,4	31,6	28,9	-11,3		2		3	18,1	16,4	16,4	0,0
	2		8	23,4	24,3	23,8	-4,9		3		6	5,9	5,6	5,6	0,0
	3		11	17,1	17,0	13,3	10,4								
	4		14	17,1	16,2	15,8	3,3								

5	17	21,4	22,7	21,1	- 8,7	4	9	17,7	17,2	17,2	- 0,0
6	20	14,5	15,4	15,3	- 1,9	5	12	25,8	24,1	24,1	- 0,0
7	23	11,9	10,8	8,5	6,7	6	15	14,9	11,5	11,5	0,0
8	26	10,4	10,3	10,3	0,9	7	18	—	4,7	4,7	0,0
9	29	13,4	13,1	12,2	- 4,9	8	21	15,3	13,1	13,1	- 0,0
3	0	3	16,7	16,7	- 0,0	9	24	16,3	15,3	15,3	- 0,0
1	6	5,6	5,6	5,6	- 0,0	1	4	11,5	10,8	10,8	6,0
2	9	16,8	16,9	16,9	0,0	2	2	14,1	11,0	11,0	3,2
3	12	22,4	24,2	24,2	0,0	3	5	18,7	16,7	16,7	- 6,1
4	15	13,1	11,7	11,7	- 0,0	4	8	13,4	13,3	13,3	- 2,4
5	18	—	4,6	4,6	- 0,0	5	11	9,8	9,7	9,7	5,7
6	21	13,0	13,0	13,0	0,0	6	14	10,9	9,5	9,5	1,7
7	24	15,3	15,3	15,3	0,0	7	17	14,6	13,2	13,2	- 4,9
4	0	4	14,8	14,3	3,6	8	20	10,3	9,3	9,3	- 1,0
1	7	18,6	15,8	14,9	5,4	9	23	7,2	6,9	6,9	3,9
2	10	12,5	9,9	9,2	- 3,8	1	2	10,6	7,7	7,7	- 2,2
3	13	10,7	9,2	8,1	- 4,4	2	1	8,9	7,5	7,5	- 3,8
4	16	13,2	12,3	11,8	3,5	3	4	10,5	10,4	10,4	2,5
5	19	13,1	11,4	10,9	3,3	4	7	11,3	10,9	10,4	3,4
5	0	5	11,5	10,9	- 3,9	5	10	7,7	7,0	6,5	- 2,6
1	8	11,3	9,5	9,3	1,8	6	13	8,1	6,6	6,0	- 2,8
2	12	8,1	6,8	5,7	3,7	7	16	7,3	7,5	7,3	2,2
3	15	7,1	6,9	6,8	1,2	8	19	8,1	7,7	7,4	2,0
4	18	8,5	8,5	7,9	- 2,9	1	3	6,1	5,9	5,9	0,0
6	0	6	2,6	2,6	- 0,0	2	0	8,2	8,1	8,1	- 0,0
1	9	5,1	4,9	4,9	0,0	3	3	5,1	4,7	4,7	0,0
2	12	7,0	7,0	7,0	0,0	4	7	—	2,2	2,2	0,0

marked by  $S_1$  and  $S_2$  in the map can be explained only by that particular coordination of S atoms, thus confirming the results obtained by JELLINEK et al. (1960). If octahedral, only one of the pair peaks should appear in the projection.

It is clear that the Mo atoms should be located at  $z = 0$ ; the rest of the  $z$ -parameters of the S atoms found in this map were refined by Fourier methods. The final parameters for the oblique cell are:

	$x'$	$z'$
Mo	0	0
$S_1$	0,752	0,257
$S_2$	0,582	0,743

Between the  $z$ -parameters of the S atoms,  $z(S_1)$  and  $z(S_2)$ , there should be a relation,  $z(S_1) = 1 - z(S_2)$ , if the  $\text{MoS}_2$  layer has a mirror plane passing through the Mo atoms. It appeared, in the course of refinement, that the above relation was not held, but showed a deviation of about 0,003. However, the deviation was reduced to less than 0,001 at a final

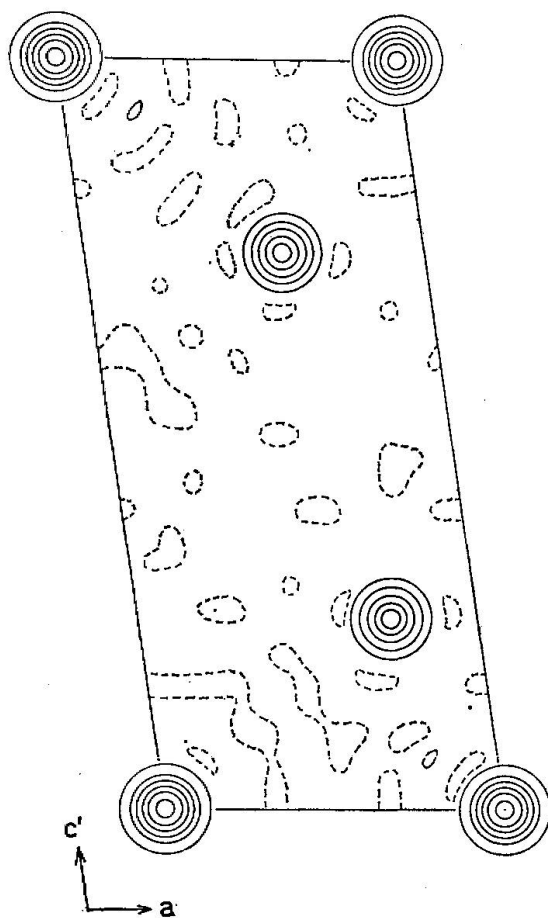


Fig. 3. Final Fourier projection of rhombohedral  $\text{MoS}_2$ . Contours are drawn at intervals of  $10 \text{ e} \cdot \text{\AA}^{-2}$  for S and  $30 \text{ e} \cdot \text{\AA}^{-2}$  for Mo, starting from  $10 \text{ e} \cdot \text{\AA}^{-2}$ . Zero contours are shown by broken lines.

stage. Since the deviation of the  $x'$  coordinates from the threefold axes were found to be slightly less than 0,001, the above deviation in the  $z$ -parameters is not significant. The errors in coordinates, thus, do not likely to exceed more than  $\pm 0,01$  for the hexagonal true cell, for which the above coordinates are expressed as:

$$\begin{aligned}\text{Mo} & \text{ at } 3(a) \text{ with } z = 0 \\ \text{S}_1 & \text{ at } 3(a) \text{ with } z = 0,247_7 \\ \text{S}_2 & \text{ at } 3(a) \text{ with } z = 0,419_0\end{aligned}$$

The final coordinates give  $R = 8,6\%$  for the  $(h0l)$  reflections. The wrong structure with octahedral coordination gives  $R = 30,4\%$ . For the atomic scattering factors, approximations by VAND et al. (1957) were used with parameters later given by FORSYTH et al. (1959), and computations were performed by a Bull-machine. The temperature coefficients  $B = 0,5$  were estimated by trial and error. The final Fourier map is shown in Fig. 3. Comparison of  $F_0$ 's and  $F_c$ 's is given in Table 1.

#### 4. Description of the structure

The MoS<sub>2</sub> layers consist of trigonal prisms of MoS<sub>6</sub> groups. The detailed configuration of the group is shown in Fig. 4a. The dimensions of the cell edges of the trigonal prism are equal within the limits of experimental errors. Bond distances and angles found for the groups are:

Mo-S	2,41 $\pm$ 0,02 Å
S <sub>1</sub> -S <sub>1</sub>	3,166 $\pm$ 0,005
S <sub>1</sub> -S <sub>2</sub>	3,15 $\pm$ 0,02
S <sub>1</sub> -Mo-S <sub>1</sub>	82,1 $\pm$ 0,5°
S <sub>1</sub> -Mo-S <sub>2</sub>	81,8 $\pm$ 0,5°

Mo-S bond distances reported for other compounds are 2,36 Å and 2,57 Å. They were found in Mo<sub>2</sub>S<sub>3</sub> by an analysis of powder data (JELLINEK, 1961). The Mo atoms in this compound have distorted octahedra of S atoms. Sulfur atoms between successive layers are at close packed positions. Distances of S-S between layers are  $3,50 \pm 0,02$  Å and slightly smaller than the sum of van der Waals radii. Each sulfur atom is surrounded by three S atoms and three Mo atoms (Fig. 4b).

The structure of rhombohedral MoS<sub>2</sub> is illustrated in Fig. 5a ( $3R$ ) and Fig. 6a ( $3R$ ). The difference from the hexagonal modification is merely in the way of layer stackings and it is best explained by the

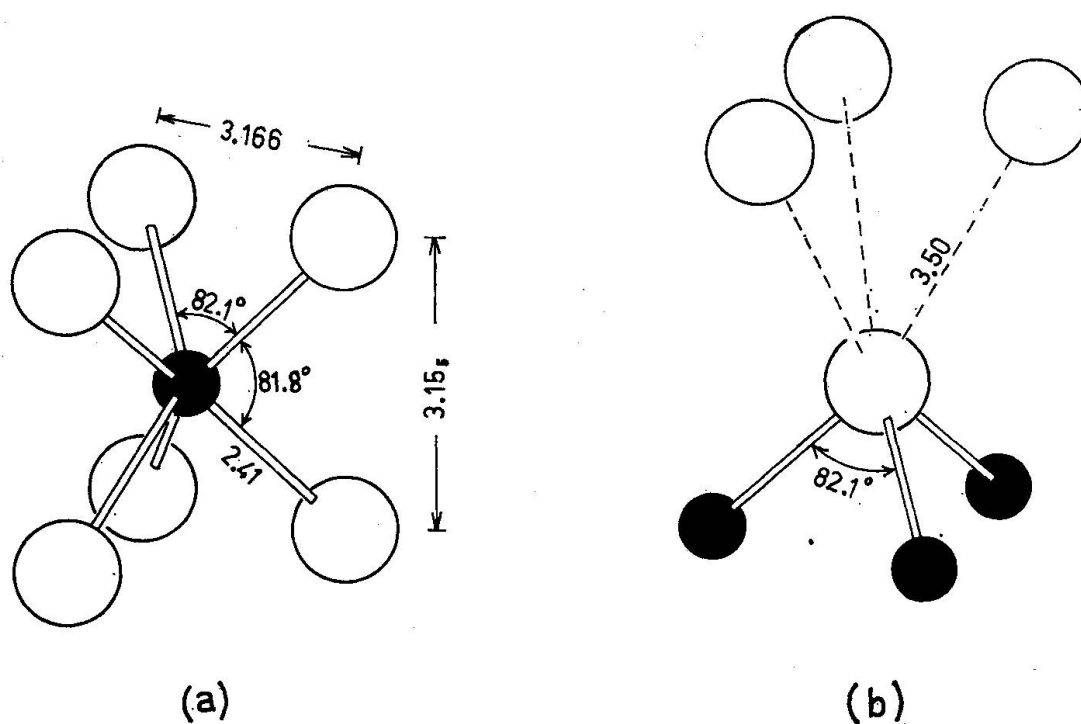


Fig. 4. (a) Configuration of  $\text{MoS}_6$  trigonal prism.  
 (b) Surroundings of sulfur atom. Mo and S are respectively shown by solid and open circles.

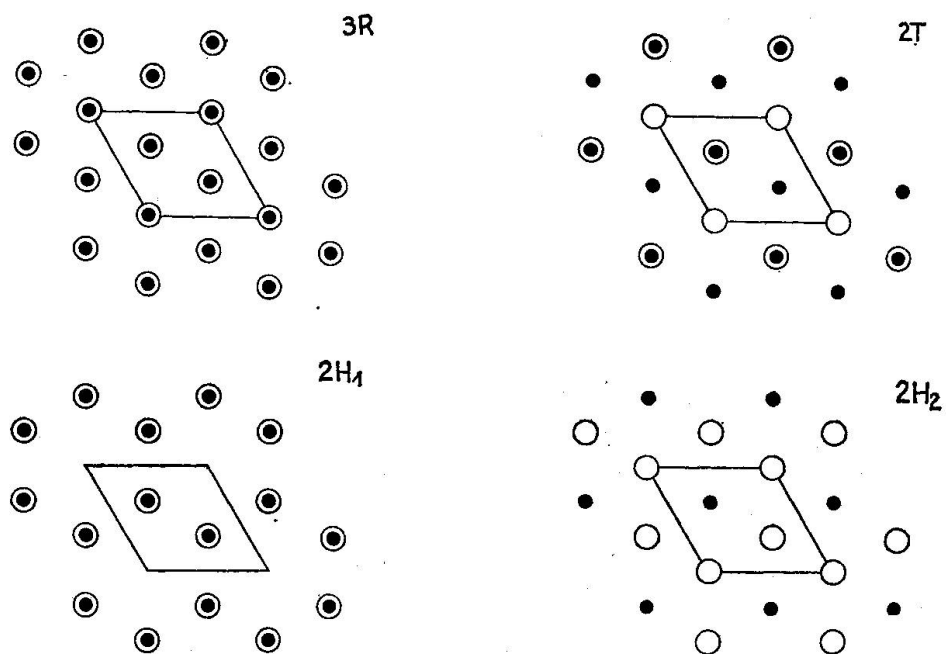


Fig. 5. Structures of simple polytypes projected along the  $c$ -axis. The origin of  $2H_2$  is not taken on a six-fold screw axis. To compare with Fig. 6, the origin of  $2H_2$  should be shifted to  $(\frac{2}{3}, \frac{1}{3}, 0)$  and that of  $2T$  to  $(\frac{1}{3}, \frac{2}{3}, 0)$ . ( $\bullet$  = Mo,  $\circ$  = S).

$a_1$ -axis projection. Both structures give the same projection along the  $[210]$  axis. This is shown in Fig. 7 together with a Fourier projection obtained by 22 ( $hhl$ ) reflections recorded on precession films.

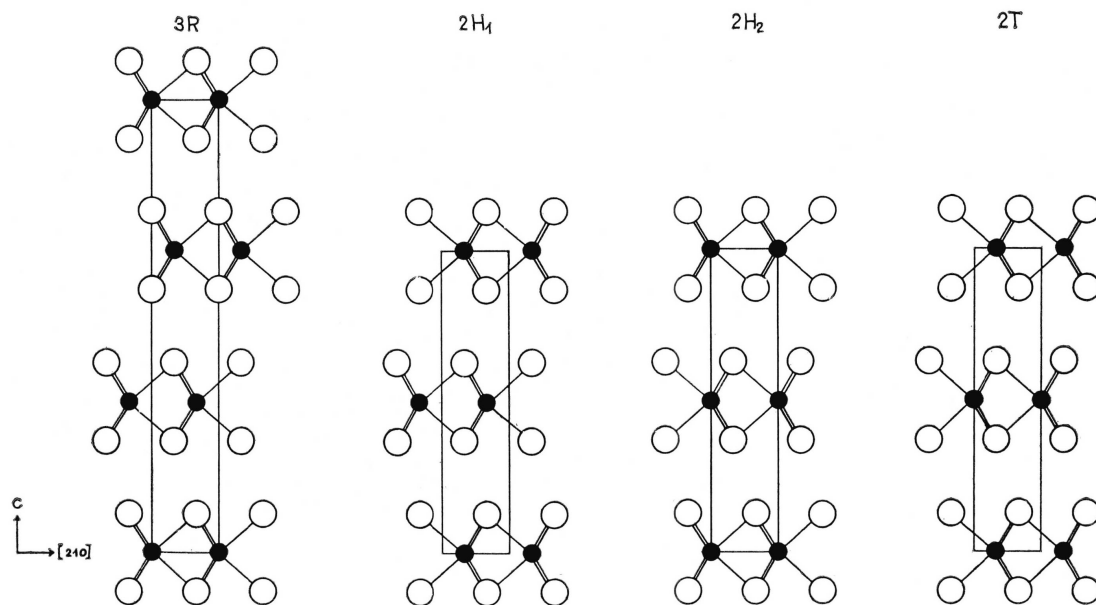


Fig. 6. Structures of simple polytypes projected along the  $a_2$ -axis.

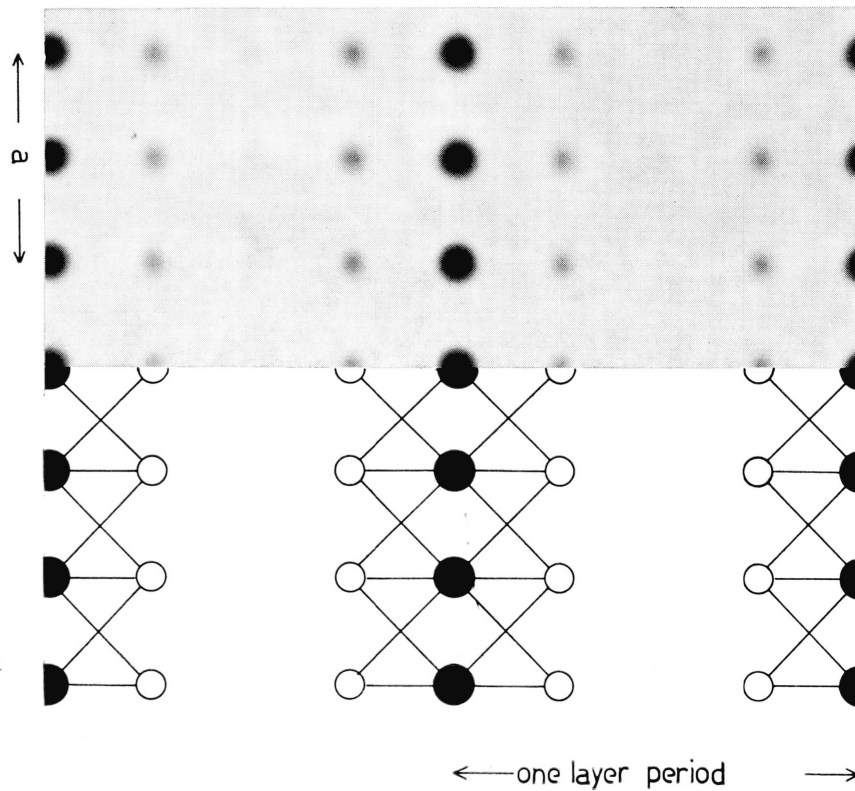


Fig. 7. Projection of rhombohedral MoS<sub>2</sub> //  $[210]$ .

## 5. Polytypes

Rhombohedral and hexagonal structures are different only in the way of stacking of the  $\text{MoS}_2$  layers. It is then suspected that there may be some other ways of stacking of these particular layers. To study this problem, a systematic derivation of possible polytypes of this structure was carried out.

The fundamental feature of the  $\text{MoS}_2$  structures is the close packing of sulfur layers between two successive  $\text{MoS}_2$  layers. A layer of  $\text{MoS}_2$  can be brought in such a close packed position by two kinds of operations, *stacking operations*, namely *translations* and *screw operations*. Possible translation operations,  $T$ , may be expressed by the form  $T_n = mt_n + t_c$ ,  $m = 1, 2 \pmod{3}$ , where  $t_n$  is the horizontal component of  $T_n$  (Fig. 8b), and  $t_c$  is the vertical component and equal to the one-layer period along the normal of the layers. As stacking operations, only primitive translations should be taken into account. Thus, the translations are expressed by  $T_n = t_n + t_c$  where  $n = 1, 2, 3$ .  $T_1$ ,  $T_2$  and  $T_3$  are equivalent by the symmetry of the  $\text{MoS}_2$  layers.

The possible screw operations are:

axis (Fig. 8b)	rotation	translation
$A$	$\alpha = \pm 60^\circ, 180^\circ$	$t_c$
$B$	$\beta = 180^\circ$	$t_c$
$C$	$\gamma = \pm 120^\circ$	$t_c$
$D$	$\delta = \pm 60^\circ, 180^\circ$	$t_c$

The structure of a polytype can be derived by a successive sequence of a stacking operation or of a combination of operations. The fundamental polytypes with ordered simple structures are those which can be derived by one kind of sequence of a stacking operation. With that condition, four polytypes are derived as follows.

### 1. Polytypes derived by translation operations

Among possible sequences of the translation operations, two are distinct. They are

(i) stacking sequence:  $T_n, \bar{T}_n, T_n, \bar{T}_n, \dots$  (Fig. 8c)

The translation sequence can be rearranged in terms of translations from starting layer to each layer. This is:

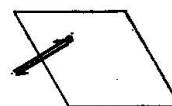
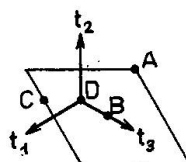
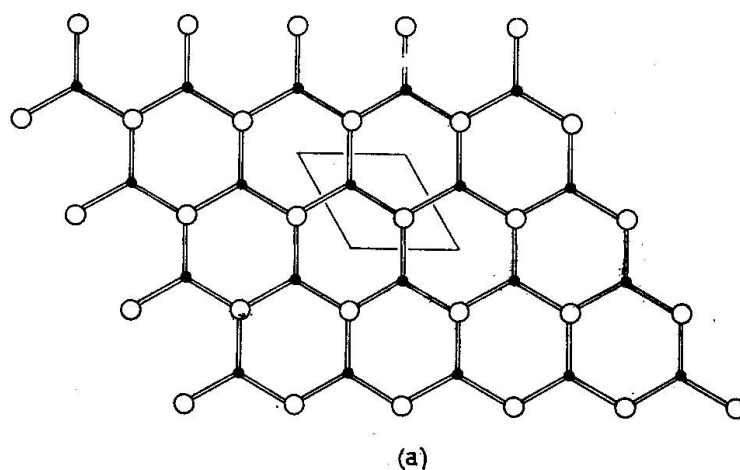
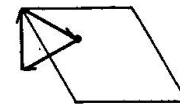
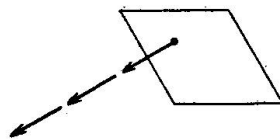


Fig. 8. One MoS<sub>2</sub> layer as seen along *c*-axis (a) and derivation of polytypes (b-e).



$$T_n, T_n \cdot \bar{T}_n, T_n \cdot \bar{T}_n \cdot T_n, \dots$$

$$= [T_n, 2t_c] \bmod 2t_c$$

These two-layer structures have a symmetry  $D_{3h}^1 - P\bar{6}m2$ .

(ii) The stacking sequence:  $T_n, T_n, T_n, \dots$  (Fig. 8d).

Similarly as above, the translations from the starting layer to each layer are

$$T_n, 2T_n, 3T_n, \dots$$

$$= [T_n, 2T_n, 3t_c] \bmod 3t_c.$$

A three-layer structure with symmetry  $C_{3v}^5 - R3m$  results.

The spiral sequence of stacking operations like  $T_1, T_2, T_3$ , gives the same rhombohedral structure (Fig. 8e).

It can be proved by similar treatment that other possible sequences of stacking operations are equivalent to one of the above two cases.

## 2. Polytypes derived by screw operations

It is evident, among possible screw operations as given above, the operation at  $B$  is equivalent to that of  $A$  with  $\alpha = 180^\circ$ . Therefore, operations at  $A, C$ , and  $D$  are the cases which should be considered. To derive polytypes by screw operations, generally, we have to study two cases, one is the case where rotation occurs alternatively and the other it occurs only in one direction (SADANAGA and TAKÉUCHI, 1961). The results obtained by a similar treatment as in the case of translations are:

screw operation at	alternative	one direction
$A$	$[A(\alpha, t_c), 2t_c] \bmod 2t_c$ $2H_1$	$[A(\alpha, t_c), 2t_c] \bmod 2t_c$ $2H_1$
$C$	$[C(\gamma, t_c), 2t_c] \bmod 2t_c$ $2T$	$[C(\gamma, t_c), C(\gamma, 2t_c), 3t_c] \bmod 3t_c$ $3R$
$D$	$[D(\delta, t_c), 2t_c] \bmod 2t_c$ $2H_2$	$[D(\delta, t_c), 2t_c] \bmod 2t_c$ $2H_2$

The polytypes derived by screw operations at  $C$  are the same with the results of translation operations. Thus, the distinct polytypes of  $\text{MoS}_2$  with simple structures are one three layer structure and three two layer structures.

The three-layer structure is rhombohedral and has space group  $C_{3v}^5 - R3m$ . Two of the two-layer structures are hexagonal and have same space group  $D_{6h}^4 - P6_3/mmc$ , but different structures. The rest of the two-layer structures has trigonal symmetry  $D_{3h}^1 - P\bar{6}m2$ . For convenience, these polytypes may be designated by  $3R, 2H_1, 2H_2$  and  $2T$  respectively just like the nomenclature of polytypes of micas (SMITH and YODER, 1956). The structure of these polytypes are compared in Fig. 5 and Fig. 6.

The  $2H_1$  structure is the ordinarily hexagonal molybdenite found in Nature. The  $3R$  whose detailed structure is described above is found in Nature and also obtained by synthesis. Other polytypes of  $\text{MoS}_2$

have not been identified. However, it should be noted that the structure proposed for hexagonal NbS<sub>2</sub> (JELLINEK, 1960) corresponds to  $2H_2$ . In view of this evidence, it may be expected that MoS<sub>2</sub> will also be obtained in that structure.

The missing rules of reflections are identical for  $2H_1$ ,  $2H_2$  and  $2T$ . The „characteristic“ missing rules for  $D_{3h}^1 - P\bar{6}m2$  and  $D_{6h}^4 - P6_3/mmc$  are different; but by „composition“ of different lattice complexes (Gitterkomplexe)  $D_{3h}^1 - P\bar{6}m2$  may give the same missing rules as  $D_{6h}^4 - P6_3/mmc$ . In the  $2T$ -structure, we have the composition of Mo<sub>1</sub> ( $\frac{1}{3}\frac{2}{3}0$ ) with Mo<sub>2</sub> ( $\frac{2}{3}\frac{1}{3}\frac{1}{2}$ ) and of S<sub>1</sub> ( $z = 0,1285, 0,8715$ ) with S<sub>2</sub> ( $z = 0,6285 = 0,1285 + \frac{1}{2}, 0,3715 = 0,8715 - \frac{1}{2}$ ). This gives the „compositional missing rule“ [Kompositionsauslöschung]:  $(hh\bar{2}hl)$  only for  $l = 2n$  present which is „characteristic“ for  $D_{6h}^4 - P6_3/mmc$  (see Int. Tabellen, v. I, p. 401, 1935, I. Laue symmetry  $D_{6h} - 6/mmm$ , A. II. 1.;  $D_{3h}^1 - P\bar{6}m2$  is a „mögliches Raumsystem“). This effect now is called „Templeton effect“ (TEMPLETON, 1956; PABST, 1959, 1961), but it is just a special case of the Kompositionsauslöschungen, recognised long ago. However, these three modifications of two-layer structures of MoS<sub>2</sub> can be effectively identified by comparing intensities of reflections. The structure factors  $F(h0l)$  are computed for these structures and compared in Table 2. To compare with the  $2T$  structure, origins of  $2H_1$  and  $2H_2$  are not taken at centers of symmetry. As seen in Fig. 5, the frame of S atoms is the same for these three structures while only the positions of Mo are different. The calculations of structure factors were made in a similar way as for the analysis of the rhombohedral modification.

If sequences of translation operations are not unique, infinite numbers of polytypes can be derived in a theoretical sense. The situation is quite similar for other layer structures like  $CdJ_2$  (MITCHEL, 1962) or SiC (RAMSDEL, 1947). For the case where the sequence  $T_n, T_n, T_n, \dots$  is interrupted by  $\bar{T}_n$  with a regular interval, various rhombohedral structures are derived. For instance, in the sequence, every third  $T_n$  is replaced by  $\bar{T}_n$ , a nine layer rhombohedral structure ( $9R$ ), and if every fourth  $T_n$  is replaced by  $\bar{T}_n$ , a  $12R$  structure will result and so on. To enumerate these kinds of complex polytypes of longer  $c$ -axis periods, to the works by RAMSDEL for SiC (1947) should be referred. The sequences of translation operations can be easily correlated with RAMSDEL's notation for SiC polytypes. Combinations of screw operations give other series of complex structures.

ZELIKMAN et al. have synthetically obtained various MoS<sub>2</sub> by different methods (1961) and showed that some of them give powder patterns

Table 2. Comparison of calculated structure factors  $F(h0l)$  of  $2H_1$ ,  $2H_2$  and  $2T$ .

$h$	$l$	$2H_1$	$2H_2$	$2T$	$h$	$l$	$2H_1$	$2H_2$	$2T$
0	4		24,2		2	0	38,2	31,1	20,3
	6		63,1			1	21,8	18,0	34,6
	8		80,9			2	22,0	46,1	23,1
	10		36,7			3	57,8	19,1	50,9
	12		13,4			4	7,4	57,8	31,5
	14		39,3			5	51,7	15,2	46,1
	16		45,6			6	22,2	38,9	19,5
						7	15,7	18,3	29,4
1	0	54,5	45,2	29,1		8	31,1	25,2	16,5
	1	31,9	25,1	49,5		9	19,6	11,5	27,2
	2	31,1	64,9	32,5		10	14,6	36,6	18,4
	3	78,3	25,1	69,3		11	43,9	15,8	38,5
	4	10,9	76,4	41,3		12	5,6	40,3	21,8
	5	66,2	18,7	59,1		13	32,7	7,8	29,6
	6	27,9	49,4	24,8					
	7	20,0	21,9	36,3	3	0		56,8	
	8	37,4	30,8	19,9		2		32,8	
	9	23,4	13,6	32,5		4		11,2	
	10	17,2	43,2	21,8		6		33,9	
	11	51,5	18,7	45,2		8		47,9	
	12	6,4	47,0	25,5		10		22,9	
	13	38,0	9,2	34,4					
	14	18,7	28,1	14,3					

These reflections are those observed on precession films by the use of Mo-radiation. The coordinates of the structures are given in Table 3.

$3R$	$x$	$y$	$z$
Mo	0	0	0
	$\frac{1}{3}$	$\frac{2}{3}$	$\frac{1}{3}$
	$\frac{2}{3}$	$\frac{1}{3}$	$\frac{2}{3}$
$S_1$	0	0	0,2477
	$\frac{1}{3}$	$\frac{2}{3}$	0,5810
	$\frac{2}{3}$	$\frac{1}{3}$	0,9143
$S_2$	0	0	0,4190
	$\frac{1}{3}$	$\frac{2}{3}$	0,7523
	$\frac{2}{3}$	$\frac{1}{3}$	0,0856
$2H_1$	$x$	$y$	$z$
Mo	$\frac{1}{3}$	$\frac{2}{3}$	$\frac{1}{3}$
	$\frac{2}{3}$	$\frac{1}{3}$	$\frac{2}{3}$
S	$\frac{1}{3}$	$\frac{2}{3}$	0,3785
	$\frac{2}{3}$	$\frac{1}{3}$	$\overline{0,3785}$
	$\frac{2}{3}$	$\frac{1}{3}$	0,8785
	$\frac{1}{3}$	$\frac{2}{3}$	0,1215

Table 3. Coordinates of Mo and S-atoms in the four deduced polytypes.

$2H_2$	$x$	$y$	$z$	$2T$	$x$	$y$	$z$
Mo	0	0	$\frac{1}{4}$	Mo <sub>1</sub>	$\frac{1}{3}$	$\frac{2}{3}$	0
	0	0	$\frac{3}{4}$	Mo <sub>2</sub>	$\frac{2}{3}$	$\frac{1}{3}$	$\frac{1}{2}$
S	$\frac{1}{3}$	$\frac{2}{3}$	0,3785	$S_1$	0	0	0,1285
	$\frac{2}{3}$	$\frac{1}{3}$	$\overline{0,3785}$		0	0	0,8715
	$\frac{2}{3}$	$\frac{1}{3}$	0,8785	$S_2$	$\frac{1}{3}$	$\frac{2}{3}$	0,6285
	$\frac{1}{3}$	$\frac{2}{3}$	0,1215		$\frac{1}{3}$	$\frac{2}{3}$	0,3715

differing from those of  $3R$  and  $2H_1$ . Though they concluded that these are the results of a disordering of  $3R$  and  $2H_1$  structures, more careful analysis is desirable to prove this hypothesis.

## 6. A note on the stacking operations of the $2T$ structure

Of the above four simple polytypes,  $3R$ ,  $2H_1$  and  $2H_2$  can be derived by continuous screw operations of a MoS<sub>2</sub> layer in one direction, while  $2T$  can not be derived in such a way. It is impressing that the stacking mechanism of layers in this polytype is completely different from those of others. However, it is possible to point out a geometrical similarity in stacking operations between  $2T$  and the hexagonal structures. Namely, the  $2T$  structure can also be obtained by a spiral sequence of translation operations (Fig. 9a) instead of alternative translations or alternating screw operations. The similarity of this stacking operations and hexagonal screw operations is shown in Fig. 9b, c.

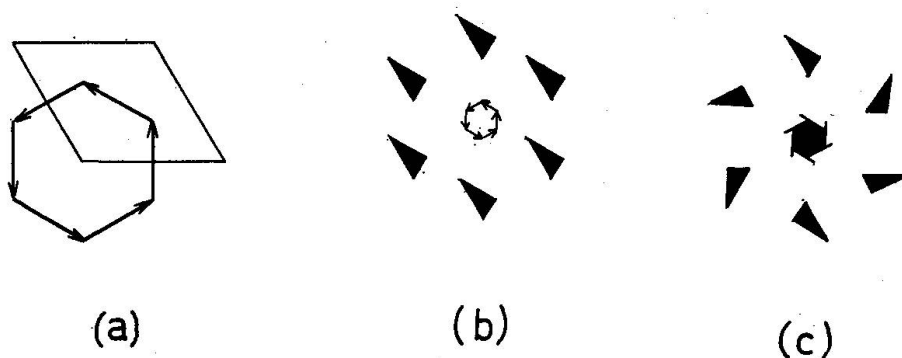


Fig. 9. Hexagonal spiral succession of translations (a). It is expressed in the similar way of screw axis (b) and compared with  $6_1$  (c).

This kind of spiral arrangement of translations is, in reality, an inherent property of the lattice structures and occurs in any crystal lattice. The point, however, which should be noted here, is that the individual translations in the stacking operations in  $2T$  are not lattice translations, but only the products of two successive translations belong to the lattice translations.

## Acknowledgements

We thank the Schweizerischer Nationalfonds and the Kommission zur Förderung der wissenschaftlichen Forschung for supporting this investigation, the Institut für Angewandte Mathematik (Prof. W. NEF and Dr. R. HÜSSER) for the

possibility of calculation on the Bull-Gamma-machine and Mr. St. GRAESER for the rhombohedral molybdenite specimen.

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Manuscript received September 5, 1963.